



## **Fabrication of flexible composite drug films via foldable linkages using electrohydrodynamic printing**

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1   **Fabrication of Flexible Composite Drug Films *via* Foldable Linkages**  
2   **Using Electrohydrodynamic Printing**

3

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## Abstract

The simple method to manufacture a flexible multi-drug with hydrophilic and hydrophobic molecules-loaded composite membrane via three dimensional (3D) electrohydrodynamic (EHD) printing has been demonstrated in this study. The composite membrane consists of two different drug-loaded sections: cellulose acetate-ibuprofen (CA-IBU) and cellulose acetate-paracetamol (CA-Para), respectively, with an intermediate polycaprolactone (PCL) folding component. The composite membranes can be folded and housed in commercial capsules to aid swallowing. By changing the number of PCL layers in the intermediate layers, it is possible to control and modify the mechanical and unfolding properties of the composite membrane. IBU and Para are loaded into the CA polymeric matrix in their amorphous states, with the matrices exhibiting Higuchi and first order release kinetics, respectively. The combination of IBU and Para can potentially be used as analgesic for patients. Magnetic nanoparticles as a functional material can be incorporated into the PCL matrix for wide targeting and traceable applications. The composite membrane here possesses good biocompatibility and flexibility; enabling extensive application prospects in drug combination therapy and personalized medicine.

*Keywords:* 3D printing; Folding; Drug delivery; Composite.

# 1. Introduction

Dosage forms administered orally (e.g. tablets, capsules, liquids) are the preferred method of giving medication and are known for patient compliance and safety [1, 2]. Despite this, limitations with respect to controlling drug release and personalizing dosing regimen remain [3]. To circumvent this several strategies have been developed including novel capsule designs (and their manufacturing) which display improved behavior in the gastrointestinal tract [4, 5]. In addition, multi-functional drug delivery devices or technologies are now coming to the fore to address challenges posed for combination therapy (e.g. for cancer and angiocardopathy [6, 7]. Recent advances have shown that both hydrophilic and hydrophobic drugs can be incorporated in a single dosage and used in synergy to achieve comprehensive therapeutic effects [8].

Three dimensional (3D) electrohydrodynamic (EHD) printing is an ambient environment, non-conventional, one-step process used to prepare 3D microstructures via pre-programmed collector movements [9]. Under optimal applied voltage, the electrified jet undergoes rapid stretching and solidification before reaching the collector. The controlled deposition of fibers is achieved by reducing the distance between nozzle and collector [10]. A variety of materials can be and have been processed to develop pharmaceutical drug carriers via this technique [11]. Compared to other 3D printing techniques (e.g. fused deposition modelling and selective laser sintering), 3D EHD printing provides a processing environment suitable for less thermal stable drugs and the ability to control the process easily and precisely [12, 13]. Moreover, mesh-structure patterning can be used to control surface hydrophobicity and thus drug release [14, 15].

Many polymers such as polyvinyl alcohol (PVA), polycaprolactone (PCL) and cellulose acetate (CA) have been used as drug carrier or matrix materials for drug dosage formulations [16-18]. PCL is a biocompatible polymer with good mechanical properties, which has been applied for various applications requiring strength, such as surgical sutures, wound dressings and filtration [19-21]. Cellulose acetate (CA) is generally prepared through an esterification reaction of cellulose with acetic anhydrides [22]. Due to the biodegradability, hydrophilic nature and excellent stability of CA, it has been widely used for many bio-interfacing applications [23-25], as drug encapsulating matrices, and for drug delivery applications [26].

1 In recent years, research has been directed towards improving comfort for patients with  
2 respect to pain when administrating treatment. Based on this, drug combination  
3 therapies have been put forward with the superiority of better analgesic effect when  
4 compared to monotherapy [27], whilst improving patient compliance and convenience  
5 [28]. Ibuprofen (IBU) is a widely used non-steroidal anti-inflammatory drug (NSAID)  
6 with very little side effects, which is used to treat pain and inflammation [29].  
7 Paracetamol (Para) is prescribed as an analgesic and antipyretic with good tolerance  
8 and safety [30]. Both IBU and Para are used for pain relief after surgery [31]. It has  
9 been proved that the combination of IBU and Para can provide a significantly more  
10 effective treatment of acute pain than monotherapies [32].

11 In this study, we demonstrate the engineering of a flexible multi-drug dosage membrane  
12 hosting both hydrophilic and hydrophobic molecules via three dimensional (3D)  
13 electrohydrodynamic (EHD) printing. The flexible composite film comprises foldable  
14 linkages and the unfolding process enables increased surface area for expedited drug  
15 release and increasing retention time which is ideal to reduce dosing. Moreover, loading  
16 into different compartments of the composite membrane is a proof-of-concept for  
17 combinatorial therapy which is shown via precision 3D printing. The flexible composite  
18 membrane with one central component can be folded and enclosed into capsules for  
19 oral administration. Composite membranes with different thickness have been  
20 fabricated by altering the number of layers of the central components (5, 10 and 20  
21 layers, respectively) to investigate the impacts of the central component layer numbers  
22 on the mechanical property and unfolding property of the composite membranes and to  
23 find the relatively best parameter. Also, the unfolding process is crucial in terms of  
24 expansion potential which is useful for the current application as it can increase the  
25 surface area of printed membranes and also prolong retention time in the  
26 gastrointestinal tract. IBU and Para were loaded into different regions of the composite  
27 membrane and were selected as model drugs to show proof-of-concept for  
28 combinatorial therapy and to enable multiple drug release behavior for personalized  
29 medicine. Magnetic nanoparticles were added into the central PCL component for  
30 potential targeting applications.

## 32 **2. Materials and Methods**

## 2.1. Materials

Cellulose acetate (CA,  $M_w = 30000$  g/mol), polycaprolactone (PCL,  $M_w = 80000$  g/mol) and ibuprofen (IBU,  $\geq 98\%$ ) were all obtained from Sigma-Aldrich, USA. Paracetamol (Para,  $\geq 99\%$ ) were purchased from Aladdin, China. Acetone, N, N-Dimethylformamide (DMF), acetic acid and phosphate buffer saline (PBS, pH= 7.4) were all obtained from Sinopharm Chemical Reagent, China. Magnetic nanoparticles ( $Fe_3O_4$  NPs) were supplied by HWRK Chem, China. Deionized water (DI water) was prepared in-house (Millipore Milli-Q Reference ultra-pure water purifier, USA). Minimum Eagle's medium (MEM, Gibco) and fetal bovine serum (FBS) were obtained from Invitrogen, USA. All chemicals used were of analytical grade.

## 2.2. Preparation of 3D EHD printing solutions

Acetone and DMF (acetone:DMF=1:1, v/v) were used to dissolve CA at the concentration of 22% (w/w) by mechanical stirring (VELP ARE heating magnetic stirrer, Italy) for 5 hours. IBU (5% w/w of CA) and Para (4% w/w of CA) were then added to the CA solution, respectively, and dissolved by mechanical stirring for 2 hours to obtain homogenous IBU-loaded CA solution and Para-loaded CA solution. The drug concentration of IBU and Para were altered to demonstrate practical feasibility of personalization and drug loading capabilities for co-therapy.

PCL was dissolved in acetic acid at a concentration of 24 %w/w under mechanical stirring (VELP ARE heating magnetic stirrer, Italy) for 6 hours at room temperature.  $Fe_3O_4$  NPs were added to the PCL solution at a concentration of 0.5 %w/w, which was dispersed through bath sonication for 2 hours to yield a homogeneous suspension.

## 2.3. Preparation of composite membrane

The 3D EHD printing set-up was shown in Fig. 1, including a stainless steel nozzle (outer diameter = 0.9mm, inner diameters = 0.7 mm), a syringe pump (KD Scientific KDS100, USA), a high voltage power supply (Glassman high voltage Inc. series FC, USA) and a controllable X-Y-Z movement stage. Firstly, the 5 ml syringe was loaded with drug solutions and infused through the printing nozzle at a flow rate of 0.3 ml/h. The path of movement stage was pre-designed and programmed (Adtech, China) in a grid pattern. Under the optimum preparation parameters (applied voltage of 2 kV,

1 nozzle-to-collector distance of 2 mm and movement stage speed of 100 mm/s), well  
2 aligned drug-loaded films with grid shapes were fabricated ( $6.5 \times 13$  mm) within 5 min.  
3 The layer number of the drug-loaded films were set as 10 layers.

4 The CA-IBU film and CA-Para film were then linked by the PCL fibrous central  
5 component to obtain a flexible, composite membrane. The distance between CA-IBU  
6 and CA-Para films was 2 mm. The printing parameters of the central component was  
7 optimized to achieve a stable printing Taylor cone. The applied voltage and nozzle-to-  
8 collector distance were set to 2 kV and 3 mm, respectively. The flow rate of PCL  
9 solution was 0.2 ml/h and the speed of the movement stage was 40 mm/s. The layer  
10 number of the central component was set to 5, 10 and 20 layers, respectively (denoted  
11 as P5, P10 and P20).

#### 12 *2.4. Mechanical characterization*

13 The mechanical characterization of tension was investigated using a universal material  
14 testing machine (Zwick/Roell Z020, Zwick, Germany). The sizes of P5, P10 and P20  
15 were all  $13.5 \times 15$  mm and the thickness of the central components were approximately  
16 40  $\mu\text{m}$ , 70  $\mu\text{m}$  and 140  $\mu\text{m}$ , respectively. Each specimen was fixed by the grips and then  
17 tensioned at a rate of 10 mm/min until the composite membranes fractured. Each  
18 sample was performed in triplicate.

#### 19 *2.5. Unfolding property test*

20 The unfolding property of the composite membranes was tested in vitro. Each  
21 composite membrane was folded and encased in empty capsules (size 0). The composite  
22 membrane-loaded capsules were then placed into a 100 ml beaker, containing 60 ml  
23 PBS (37 °C). After being left to stand for 15 minutes, the unfolding angles of composite  
24 membranes with different folding-part layers were measured using ImageJ software  
25 (version 1.48, National Institute of Health, USA). The same test but placed in a HZ-  
26 8801K thermostatic oscillator (Taicang Science and Education Factory, China) was  
27 also performed and analyzed.

#### 28 *2.6. Fourier transform infrared (FTIR) spectroscopy*

29 The spectra of CA, Para, PCL and the composite membrane were analyzed by Fourier  
30 transform infrared spectroscopy (FTIR, IR Affinity 1, Shimadzu, Japan). For each test

disk, 3 mg sample was mixed with 300 mg KBr powders and then pressed into a transparent disk. The disks were then used for FTIR analysis between 400 and 4800  $\text{cm}^{-1}$  and the resolution was set as 4  $\text{cm}^{-1}$ .

#### 2.7. Differential scanning calorimetry (DSC) and X-ray diffraction (XRD)

Thermal analysis was performed using differential scanning calorimetry (DSC) (PE DSC 7, Perkin-Elmer, USA). Samples (~3 mg) were heated over a temperature range of 30 to 260  $^{\circ}\text{C}$ , at a heating rate of 10  $^{\circ}\text{C}/\text{min}$  in the  $\text{N}_2$  environment.

XRD analysis of each sample was performed by X-ray crystal diffractometer (Gemini A Ohra, Oxford, UK). The printed membrane was cut into pieces and then sent to XRD test center for analysis. The scan range of 2 theta was between 5  $^{\circ}$  and 60  $^{\circ}$  at a step size of 0.02  $^{\circ}$ . The voltage and intensity were 40 kV and 40 mA.

#### 2.8. In vitro drug release

HCl solution (36.46 g/mol) and phosphate buffered saline (PBS, pH=7.4) were used to prepare simulated gastric fluid (SGF, pH=1.7) and simulated intestinal fluid (SIF, pH=7.4), respectively. Drug release tests were performed using a HZ-8801K thermostatic oscillator (Taicang Science and Education Factory, China) at 50 rpm and at the ambient temperature of  $37 \pm 0.5$   $^{\circ}\text{C}$ . The tests were first performed in 20 ml SGF for 2 hours and then in 20 ml SIF for 6 hours. The characteristic UV peaks of IBU and Para were 222 nm and 243 nm, respectively (UV-2600 spectrophotometer, Shimadzu, Japan). At time intervals, 3 ml of the release solution was extracted for UV analysis, and then same volume of fresh medium was supplied to the release solution. Each sample was tested in triplicate. The cumulative drug release percentage can be estimated as Equation 1 [18].

$$\text{Drug release (\%)} = \frac{C_t}{C_M} \times 100\% \quad (1)$$

Where  $C_t$  is the drug release in the solution at time  $t$  and  $C_M$  is the total drug release in the solution.

The drug encapsulation efficiency (EE) can be calculated as Equation 2 [33].

$$\text{EE (\%)} = \frac{\text{Amount of drug encapsulated in membranes}}{\text{Theoretical amount of drug}} \times 100\% \quad (2)$$



1 The drug loading capacity (LC) can be obtained using Equation 3 [34].

2 
$$LC (\%) = \frac{\text{Amount of drug encapsulated in membranes}}{\text{weight of the membrane}} \times 100\% \quad (3)$$

3 *2.9 Kinetics of drug release*

4 The drug release mechanisms of IBU and Para from composite membranes were  
5 analyzed by fitting release data to zero-order, first-order, Higuchi and Korsmayer–  
6 Peppas mathematic models [14, 35]. The best fit model for drug release can be  
7 determined by comparing the correlation coefficient ( $R^2$ ).

8 *2.10. Scanning electron microscopy (SEM) characterization*

9 Scanning electron microscopy (SEM, Eindhoven, Netherlands) was used to investigate  
10 the morphology of the composite membranes. Each Sample was sputter-coated with  
11 gold for 90 s before the observation. The accelerating voltage was set as 10 kV. ImageJ  
12 software (National Institute of Health, USA) was used to determine the diameters of  
13 the printed fibers. The diameter distribution diagrams were obtained by Origin software  
14 (version 8.0, Originlab, USA).

15 *2.11. Magnetic test*

16 The magnetic property of the printed composite membrane was investigated using a  
17 vibration sample magnetometer (VSM Mini-CFM measurement system, Cryogenic  
18 Ltd., UK) at 300 K. Composite membrane with a known weight was placed in the  
19 magnetometer and measured in a range of -10000 Oe to 10000 Oe.

20 *2.12. Cell culture*

21 In order to analyze the biocompatibility of the printed composite membrane, L929  
22 mouse fibroblast cells were cultured on the composite membrane with MEM medium  
23 containing 1% antibiotic-antimycotic and 10% FBS in a 6 cm diameter culture dish at  
24 37 °C, under 5% CO<sub>2</sub> environment.

25 After 3 days incubation, cells were fixed with 4 (v/v) % formalin for 30 min and then  
26 permeabilized by 0.1% Triton X-100 for 5 min. After being washed three times with  
27 PBS, Alexar Fluor 546 phalloidin (1:200 v/v of phalloidin:PBS, Invitrogen, USA) were  
28 used to stain the cell cytoskeleton for 20 min followed by triple PBS rinses. 4', 6'-

diamidino-2-phenylindole hydrochloride (DAPI, 1:5000 v/v of DAPI:PBS, Invitrogen, USA) was then used to stain the cell nuclei for 5 min followed by PBS rinses two times. The cell distribution on the composite membrane was observed by a fluorescent microscope (Nikon Ti-S, Japan).

### 3. Results and Discussion

#### 3.1. Preparation and characterization of composite membranes

The printed composite membranes and their capsules are shown in Fig. 2. The composite membranes demonstrated grid structure as a result of the pre-determined patterns. The central component (PCL) provides flexibility; allowing the composite membrane to fold and thus allowing the composite membranes to be enclosed in size 0 capsules for oral administration. To investigate the effect of the central component layer numbers on mechanical properties and unfolding properties, composite membranes with 5 layers, 10 layers and 20 layers of intermediate PCL component, respectively, were printed and denoted as P5, P10 and P20 accordingly, and the mean weights were measured to be  $3.2 \pm 0.1$ ,  $3.6 \pm 0.2$  and  $4.2 \pm 0.1$  mg, respectively.

The tensile stress-strain curves of the composite membranes with different folding layers (P5, P10 and P20) are shown in Fig. 3a. It can be seen that the stress-strain curve of P5 was typical for fiber tension exhibiting three characteristic stages including a linear elastic section (region A), a progressive deformation section (region B) and a failure section (region C) [36]. It has been proven that thickness of the fibrous mats affects the its strength [37]. The tensile stress-strain curve of P10 was similar to that of P5 with three characteristic stages. However, here there was a much larger maximum stress due to the increased number of layers of PCL in P10.

The stress-strain curve of P20 showed very different results. An arrow-like pattern with a much smaller elongation was found. During the tension process of P20, the fiber breakage happened on the joint between the drug-loaded sectors and the folding sector, as the insert in Fig. 3a shows.

As shown in Fig. 3b, as the layer number of the central component increased, the

1 maximum stress of the composite membrane increased from  $0.47 \pm 0.01$  Mpa to  $1.52 \pm$   
2  $0.20$  MPa, while the maximum strain of the composite membrane decreased from  $182$   
3  $\pm 35\%$  to  $15 \pm 3\%$ . The elastic moduli of P5, P10 and P20 are calculated to be  $0.07$ ,  
4  $0.36$  and  $0.72$  Mpa, respectively. It is evident that the number of PCL layers plays an  
5 important role in the mechanical properties of the composite membranes. The good  
6 mechanical property is beneficial for the unfolding property and resisting destruction  
7 ability of composite membranes.

8 To investigate the impact of the central component layer numbers on the unfolding  
9 properties, unfolding tests of the composite membranes (P5, P10 and P20) were  
10 performed with the unfolding angle data being depicted as shown in Fig. 4. After 15  
11 minutes' standing, the capsule shells dissolved, and the folded composite membranes  
12 re-spread out as the inserts of Fig. 4a shows. The unfolding angles of P5, P10 and P20  
13 were calculated to be  $44.2 \pm 5.8^\circ$ ,  $116.0 \pm 5.4^\circ$  and  $145.2 \pm 2.2^\circ$ , respectively. This  
14 indicates that the unfolding properties of the composite membranes improved with the  
15 increase of the central component layer numbers. The results of the similar unfolding  
16 test placed in a HZ-8801K thermostatic oscillator for 15 minutes, showed the same  
17 varying tendency that the unfolding angles changed from  $92.2 \pm 7.4^\circ$  to  $148.7 \pm 18.3^\circ$ ,  
18 as the central component layer number increased from 5 layers to 10 layers. In addition,  
19 the unfolding angles of composite membranes with oscillation were larger than those  
20 without oscillation, especially P5 which was due to the smaller residual strain and better  
21 flexibility. This indicates that oscillation can facilitate the unfolding process of the  
22 composite membranes. Taking into consideration of the mechanical properties and  
23 unfolding properties as well as the preparation process, P10 was selected as a sample  
24 for subsequent studies.

25 The SEM images of three all 3 sectors in the composite membrane are shown in Fig. 5.  
26 Fig. 5a shows the micrograph of the central component with a grid structure. The  
27 diameter distribution of the PCL fibers is between  $4 \mu\text{m}$  to  $19 \mu\text{m}$ , and the mean  
28 diameter of PCL fibers is  $10.5 \pm 3.2 \mu\text{m}$ . Fig. 5a' depicts a detailed image of a PCL  
29 fiber, showing the precise stack of the printed PCL fibers with 10 layers. The SEM  
30 images of the CA-Para sector and CA-IBU sector of the composite membrane are  
31 shown in Figs. 5b and 5c, respectively. It can be seen that both CA-Para matrix and  
32 CA-IBU matrix were designed and printed in a grid structure with the grid cell size of

1  $500 \times 500 \mu\text{m}^2$ . The mean diameters of CA-Para fibers and CA-IBU fibers are  $15.7 \pm$   
2  $3.6 \mu\text{m}$  and  $18.4 \pm 4.8 \mu\text{m}$ , respectively. As shown in Figs. 5b' and 5c', both printed  
3 fibers were well aligned and stacked layer by layer precisely with uniform  
4 morphologies. This demonstrates the precision control via 3D EHD printing to  
5 manufacture patterned composite membranes.

### 6 *3.2. FTIR analysis of the composite membrane*

7 To confirm the stability of the compositions and the possible interactions between the  
8 materials of the composite membrane, FTIR analysis was performed (Fig. 6). For pure  
9 PCL, the characteristic absorption peaks at  $2947 \text{ cm}^{-1}$  and  $1730 \text{ cm}^{-1}$  are attributed to  
10 the asymmetric stretching of  $\text{CH}_2$  and carbonyl stretching, respectively [38]. The  
11 absorption peak at  $1170 \text{ cm}^{-1}$  is related to the stretching of  $-\text{C}-\text{O}$  [39]. These peaks can  
12 be identified in the composite membrane spectrum, confirming the presence of PCL in  
13 the composite membrane.

14 For pure CA, the characteristic bands at  $3495 \text{ cm}^{-1}$ ,  $1373 \text{ cm}^{-1}$  and  $1045 \text{ cm}^{-1}$  are due to  
15 the  $-\text{O}-\text{H}$  group stretching,  $-\text{C}-\text{CH}_3$  group stretching and  $-\text{C}-\text{O}-$  bond stretching,  
16 respectively [40]. For pure IBU, the intense peak at  $1720 \text{ cm}^{-1}$  is assigned to the  
17 carbonyl ( $\text{C}=\text{O}$ ) stretching [41], which is overlapping with PCL peak and can be seen  
18 in the composite membrane spectrum. The absorption bands at  $1230 \text{ cm}^{-1}$  and  $937 \text{ cm}^{-1}$   
19 are due to the stretching of  $\text{C}-\text{C}$  bond and rocking vibration of  $\text{CH}_3$  [42], respectively,  
20 which, although weak, can be found in the spectrum for composite membrane. For pure  
21 Para, the characteristic absorption peaks located at  $3319 \text{ cm}^{-1}$  and  $1554 \text{ cm}^{-1}$  are  
22 corresponding to  $-\text{O}-\text{H}$  group stretching and  $\text{N}-\text{H}$  group stretching, respectively [27].  
23 All these characteristic peaks appear in the composite membrane spectrum, suggesting  
24 the presence of the drugs in the printed composite membrane. Hence, no additional  
25 peaks or peak changes were observed in the FTIR spectrum, indicating no physical  
26 incompatibilities between the components.

### 27 *3.3. DSC and XRD analysis of the composite membrane*

28 The DSC curves of pure PCL, CA, Para, IBU and the printed composite membrane are  
29 shown in Fig. 7a. An endothermal peak is present at  $56.7^\circ\text{C}$  for pure PCL, indicating  
30 the melting point ( $T_m$ ) of PCL [42]. For pure CA, the first endothermal peak at  $67.5^\circ\text{C}$   
31 and the second peak observed at  $230.9^\circ\text{C}$  correspond to the glass transition temperature

( $T_g$ ) of CA, and the melting peak of CA is very close to the its  $T_g$  peak around 237 °C [43]. The DSC curve of pure Para shows a well-defined melting peak at 169.2 °C [3]. The sharp endothermic peak located at 77.2 °C corresponds to the melting temperature of pure IBU [44]. For the composite membrane, there are no significant changes in the melting peaks of the polymer PCL and CA, but a slight shift can be identified from 56.7 °C to 51.1 °C for PCL and from 230.9 °C to 222.8 °C for CA. Besides, the endothermal peaks of Para and IBU cannot be seen in the DSC curve of the composite membrane, indicating the amorphous state of both drugs in the printed composite membrane.

The XRD results further confirmed that both drugs were incorporated in the composite membrane in their amorphous forms. As shown in Fig. 7b, the sharp crystalline diffraction peaks of pure Para and IBU were not detectable in the XRD spectrum of the composite membrane because of their dispersion within the matrix material[18]. Although the drug loading was low, it was above the detection limits and can be successfully detected [12, 45, 46].

#### 3.4. *In vitro drug release test*

The release studies were performed in an acidic medium (gastric pH) for the first two hours then at intestinal pH (PBS) for 6 hours. As seen in Fig. 8, almost 100% of Para was released in the first 2 hours in the gastric medium. In contrast to Para release, IBU release was slower: almost 35% IBU was released during the gastric phase with subsequent sustained release in the intestinal phase. These different release behaviors may be due to the fact that Para is hydrophilic, and IBU is hydrophobic and weak acidic with poor solubility in the acidic media [47, 48]. Fig. 9 shows the SEM images of the three parts of the composite membranes after 8 h drug release test. It can be seen that CA-Para, central and CA-IBU components of composite membrane maintain their grid structures and show no obvious eroding or disintegrating. The magnified images in Figs. 9a', 9b' and 9c' show that the surface morphologies of CA-Para, CA-IBU and central components were all uniform after drug release due to high polymer stability.

The release mechanisms of Para and IBU from the composite membrane was investigated by fitting the release data to zero-order model, first-order model, Higuchi model and Korsmayer–Peppas mathematic model, as shown in Tab. 1. For Para release, it is best fitted by first order model with the highest correlation coefficient  $R^2 = 0.9759$  [49]. IBU was best fitted by Higuchi kinetics with  $R^2$  value of 0.9928, indicating the

1 Fickian diffusion mechanism of IBU, which is further confirmed by Korsmeyer-Peppas  
2 modelling with n value of 0.5 [50].

3 The EE of IBU and Para were calculated to be  $85.7 \pm 7.2\%$  and  $87.9 \pm 1.8\%$ ,  
4 respectively. Maximum EE is not obtained because drug partially diffuses from the  
5 semi-liquid polymer on the collector due to evaporation [51]. Moreover, the LC of IBU  
6 and Para were calculated to be  $2.5 \pm 0.3\%$  and  $1.3 \pm 0.2\%$ , respectively. The results  
7 indicate the feasibility of encapsulating drugs via 3D EHD printing.

### 8 *3.5. Magnetic property*

9 Magnetic nanoparticles ( $\text{Fe}_3\text{O}_4$  NPs) could be added in the central component, by  
10 simply adding them to the printing solution, to further functionalize the composite  
11 membranes with magnetic properties. Vibration sample magnetometer (VSM) analysis  
12 was performed to quantify the magnetic properties of the composite membrane at a  
13 constant temperature (300 K). The hysteresis loop of the printed composite membrane  
14 in the field range between -10000 Oe and 10000 Oe is shown in Fig. 10a. The hysteresis  
15 loop shows a S-shaped curve with no remanence and coercivity, and the curve goes  
16 through the zero point. The saturation magnetization value of the composite membrane  
17 is 0.16 emu/g. Therefore, the results manifest the superparamagnetic nature of the  
18 printed composite membrane [52]. For better visualization, the magnetic property of  
19 the composite membrane was evaluated by a permanent magnet as shown in Fig. 10b.  
20 The composite membranes (in distilled water) were drawn to the attracting side by a  
21 magnet and was recovered after removal of the magnet. The magnetic composite  
22 membranes may have many potential applications such as (but not limited to) magnetic  
23 targeting, visualization in MRI and magnetic heating [53, 54].

### 24 *3.6. Biocompatibility analysis*

25 PCL and CA are FDA approved materials which are widely used in the pharmaceutical  
26 industry [55, 56]. However, it is still necessary to investigate the biocompatibility of  
27 the composite membranes, due to the potential toxicity of actives and solvents used  
28 during the preparation procedures. Herein, L929 cells were cultured on the composite  
29 membrane for 3 days. The morphologies of L929 cells were observed through  
30 fluorescence microscope as shown in Fig. 11. Figs. 11a, 11b and 11c display the merged  
31 fluorescent images of cells growing on the CA-IBU part, CA-Para part and the central

component, respectively. Figs. 11a', 11b' and 11c' are the magnified images of Figs. 11a, 11b and 11c, respectively, showing the detailed cell morphology on the composite membrane. From the inserts of bright field images, it can be seen that cells grew well with the composite membranes, highlighting the good biocompatibility of the composite membranes.

#### **4. Conclusion**

In summary, multi-drug-loaded composite membranes consisting of the drug-loaded parts and intermediate folding section were prepared via 3D EHD printing successfully. The composite membrane found to deliver two drugs with well-defined drug release mechanisms. IBU and Para were respectively loaded in two different CA films in their amorphous forms, and linked by the PCL folding layer for the treatment of fever or pain alleviation. The mechanical property and unfolding property of the composite membranes were controllable by regulating the layer number of the folding layer. The composite membranes were able to be folded and capsulated into capsules for ease of swallowing, and re-spread out during drug release process. The magnetic property of the composite membrane can be obtained by adding magnetic NPs into the middle PCL layer for targeting applications. The good biocompatibility of the composite membrane was proved via L929 cell culture experiments. These composite membranes were controllable and flexible, and possess potential applications in drug combination therapy and personalized medicine.

#### **Acknowledgements**

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## 2 **Table and Figure Captions**

### 3 **Tables**

4 **Table 1.** Fitting parameters of Para and IBU release from the composite membrane.

5

### 6 **Figures**

7 **Figure 1.** (a) The schematic of the 3D EHD printing set-up. (b) Image of stable printing  
8 cone. (c) The design of the composite membrane, consisting of two drug loaded parts  
9 (CA-IBU and CA-Para) and a central component (PCL-Fe<sub>3</sub>O<sub>4</sub> NPs).

10 **Figure 2.** Photographs of printed composite membranes with different layers of folding  
11 part (P5, P10 and P20) with a scale of 10 mm.

12 **Figure 3.** (a) Stress-strain curves of different printed composite membranes (P5, P10  
13 and P20), the inserts show the images of P5, P10 and P20 after tension tests with a scale  
14 of 10 mm. (b) Maximum stress and Maximum strain of P5, P10 and P20.

15 **Figure 4.** The unfolding study of different composite membranes (P5, P10 and P20):  
16 (a) unfolding angles and images of P5, P10 and P20 after 15 minutes' standing; (b)  
17 unfolding angles and images of P5, P10 and P20 after 15 minutes' oscillation.

18 **Figure 5.** SEM images of the composite membrane and the diameter distribution  
19 graphs: (a) Central component with 10 layers; (b) CA-Para part with 10 layers; (c) CA-  
20 IBU part with 10 layers. (a'), (b') and (c') are magnified images of (a), (b) and (c),  
21 respectively.

22 **Figure 6.** FTIR spectra of pure PCL, CA, Para, IBU and the composite membrane.

23 **Figure 7.** (a) DSC curves of pure PCL, CA, Para, IBU and the composite membrane.  
24 (b) XRD patterns of pure PCL, CA, Para, IBU and the composite membrane.

25 **Figure 8.** Drug release profiles of Para and IBU from the composite membrane in the  
26 simulated gastrointestinal tract.

27 **Figure 9.** SEM images of the composite membrane after 8h drug release: (a) central

1 component; (b) CA-Para part; (c) CA-IBU part. (a'), (b') and (c') are magnified images  
2 of (a), (b) and (c), respectively.

3 **Figure 10.** (a) Magnetic curves of the composite membrane. (b) Composite membranes  
4 response to an externally applied magnetic source.

5 **Figure 11.** Fluorescent micrographs of L929 cells with the composite membrane: (a)  
6 CA-IBU part; (b) CA-Para part; (c) Central component. (a'), (b') and (c') are magnified  
7 images of (a), (b) and (c), respectively. Inserts in (a'), (b') and (c') are bright field  
8 images.

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**Tables**

**Table 1**

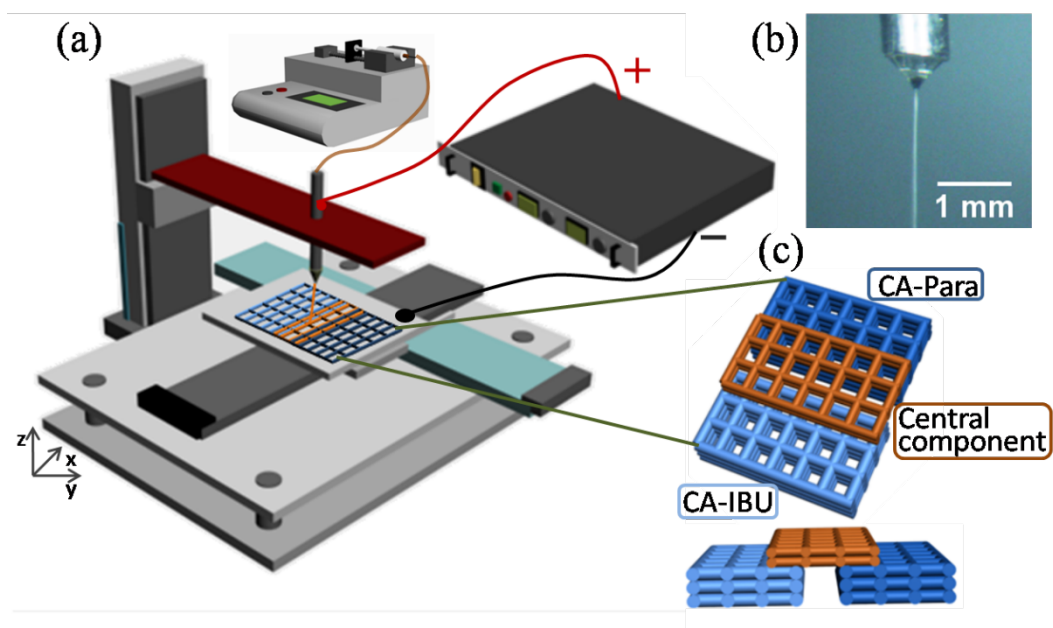
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Para	0.4407	0.9759	0.7621	-	-
IBU	0.9195	0.9701	0.9928	0.9885	0.5

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4 **Figures**



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**Figure 1**

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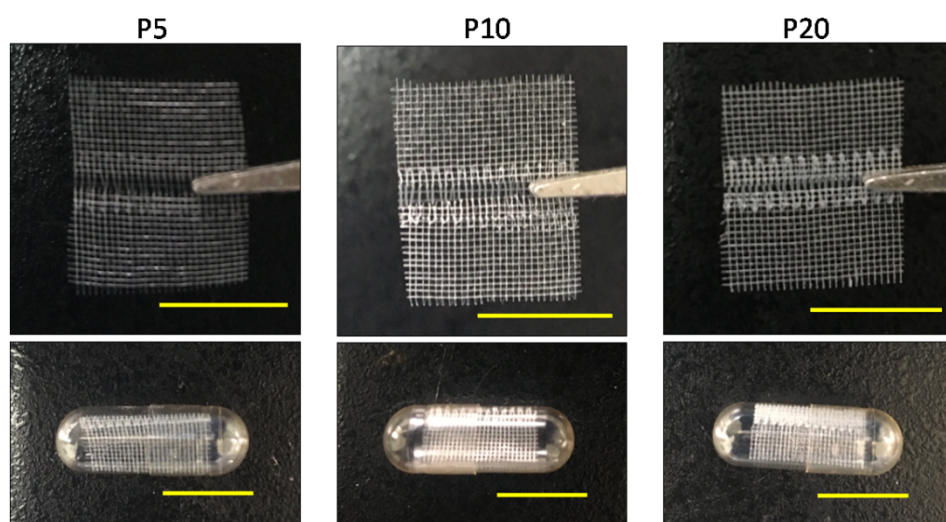
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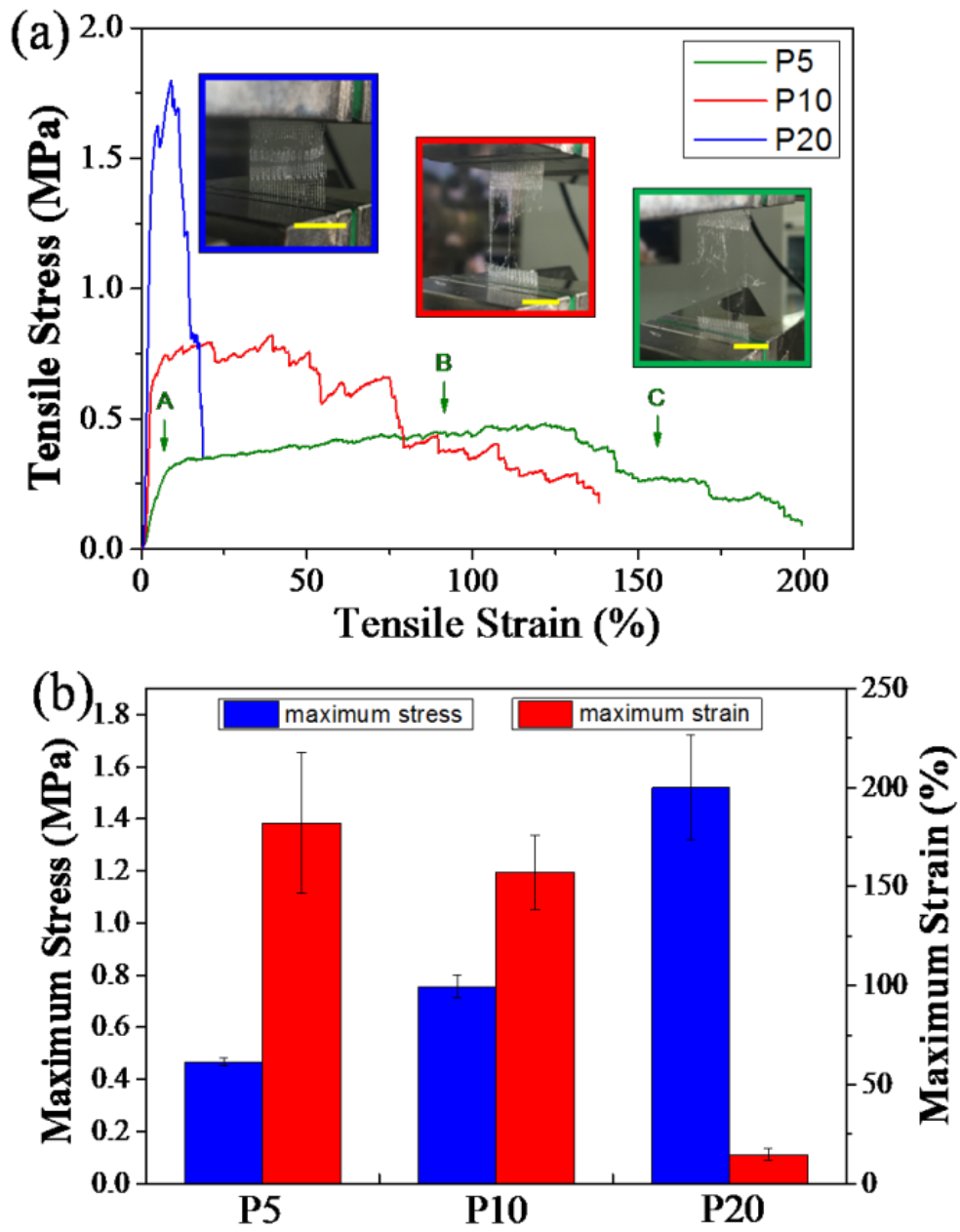
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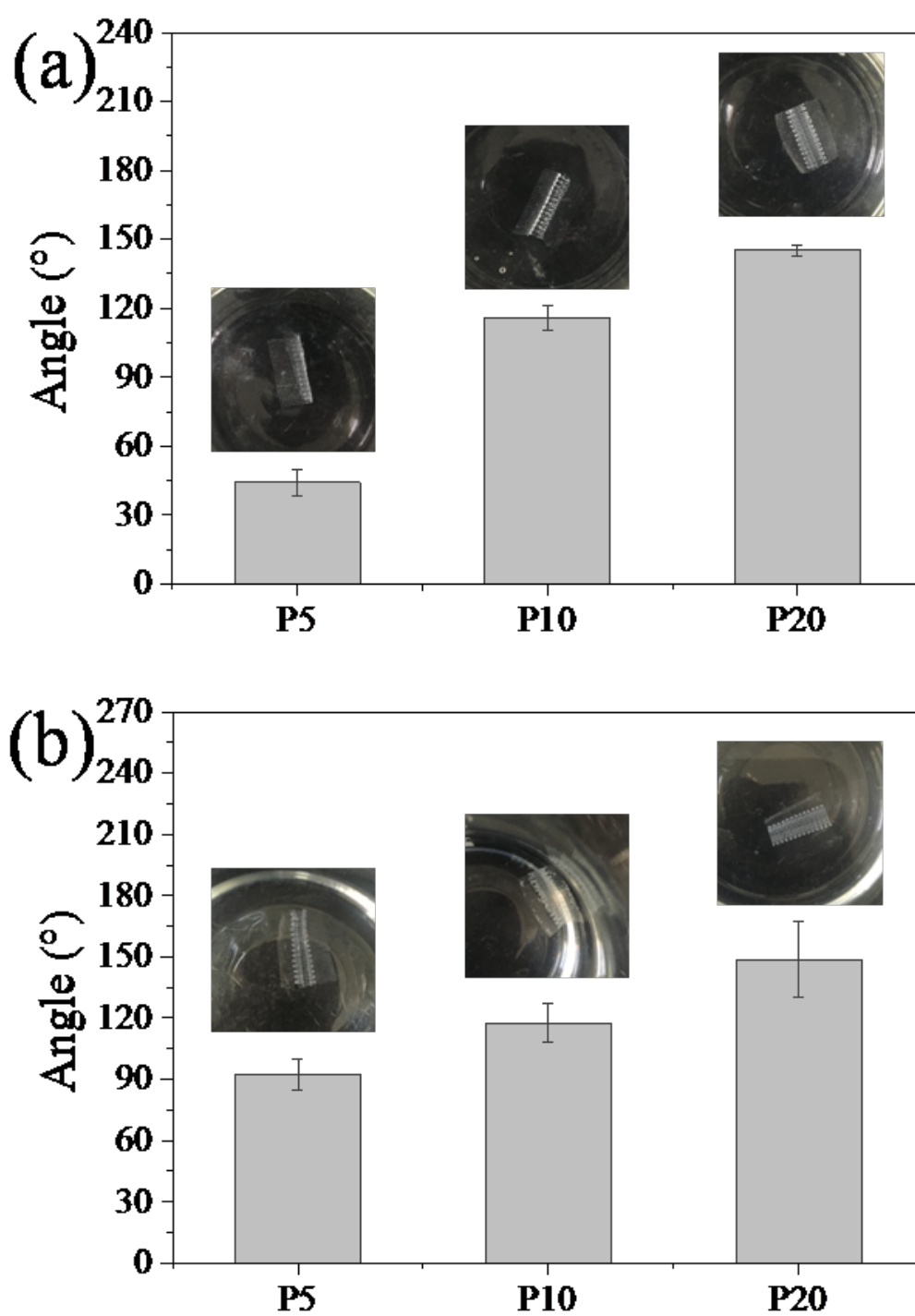


Figure 4

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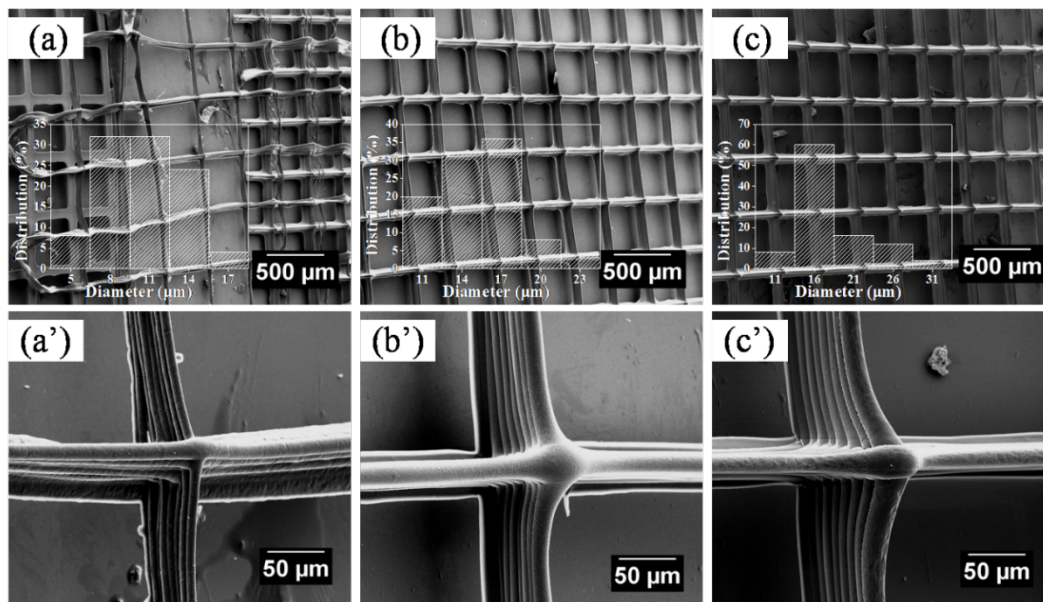
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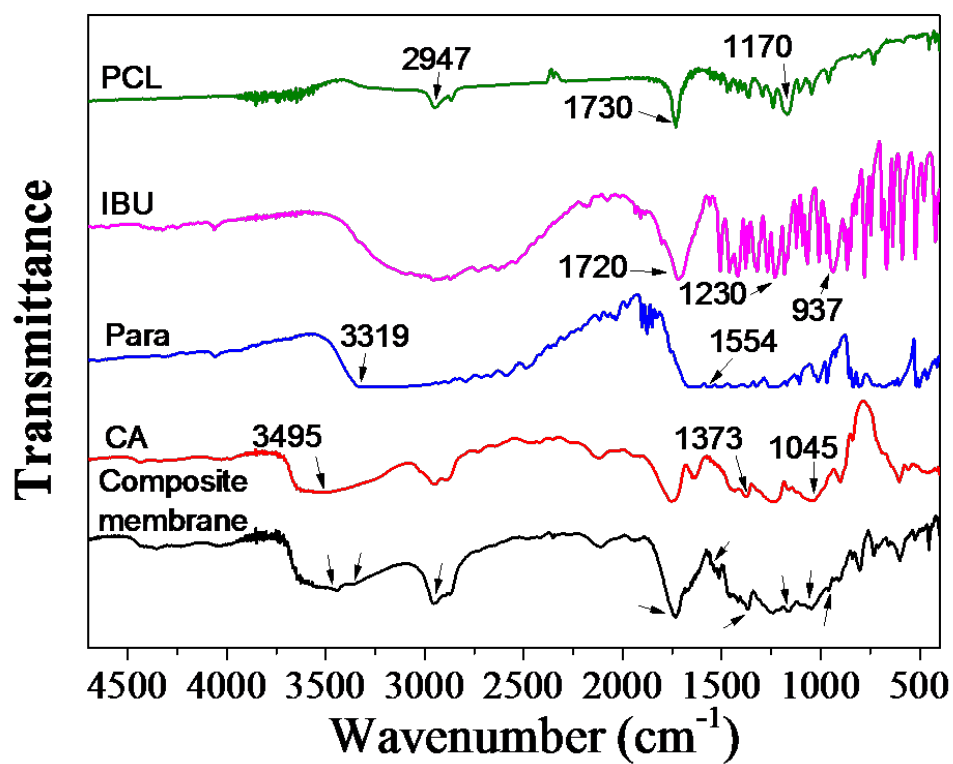
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Figure 6

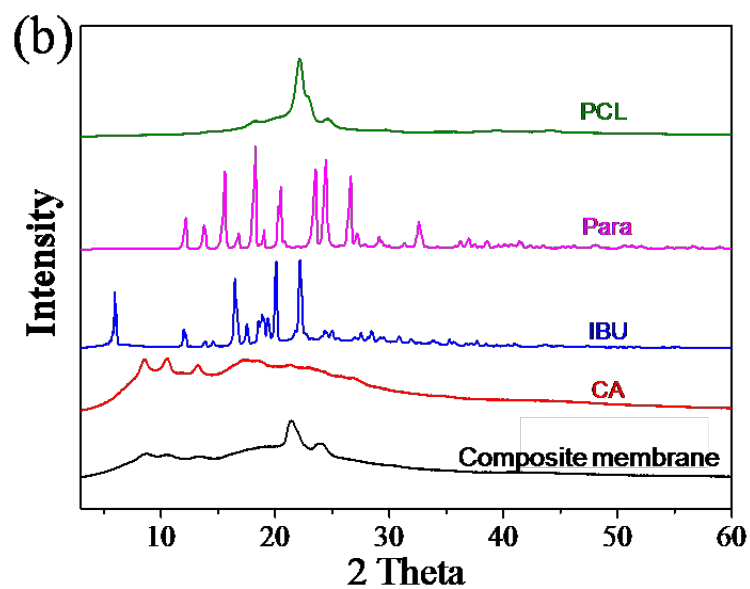
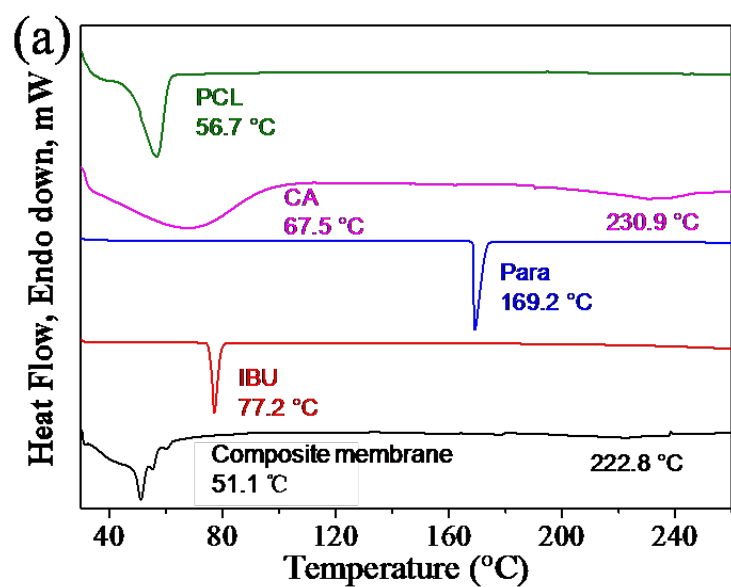
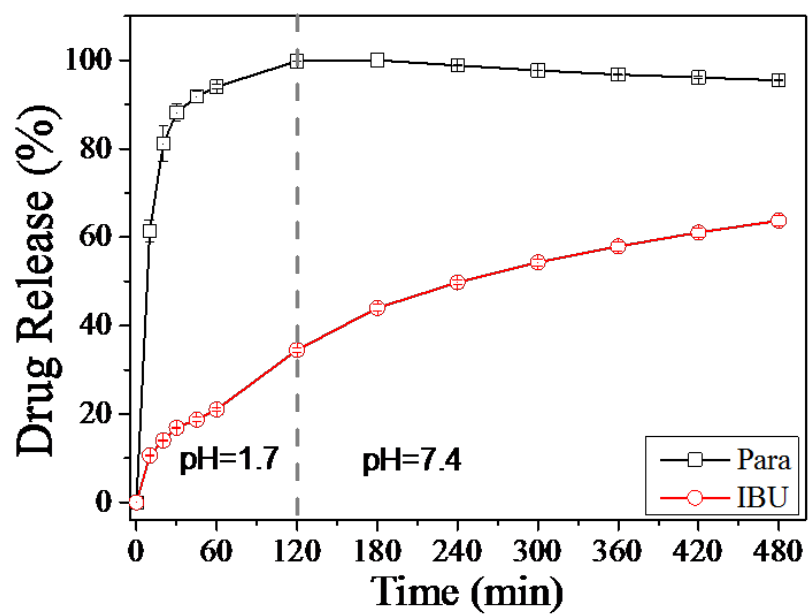


Figure 7

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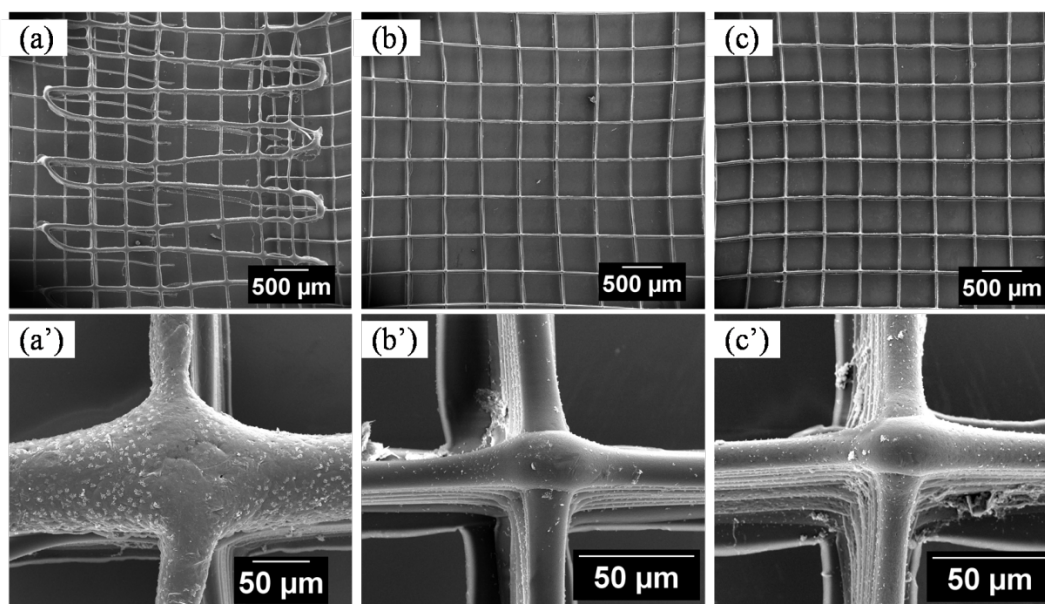
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**Figure 9**

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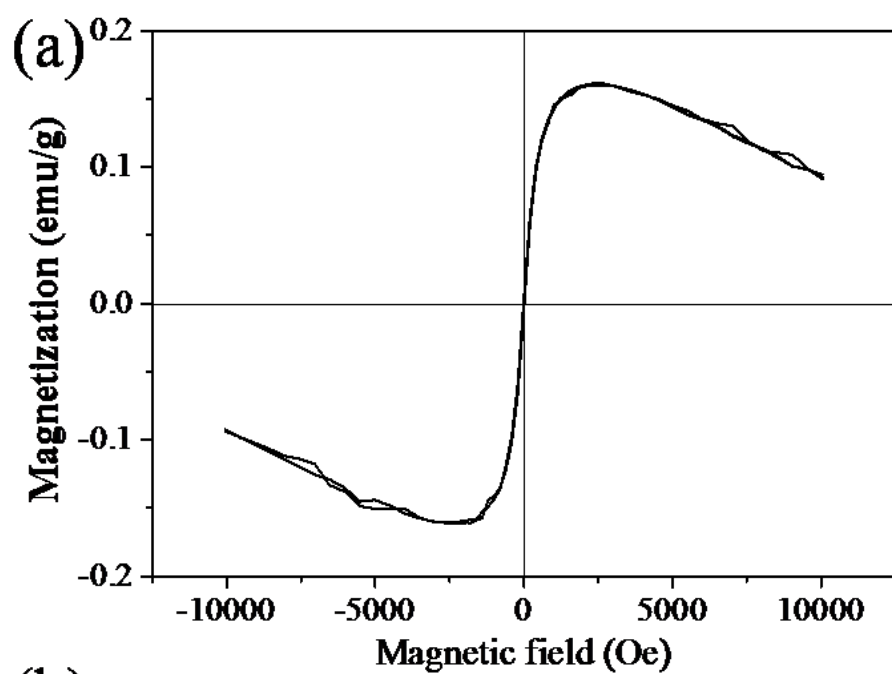
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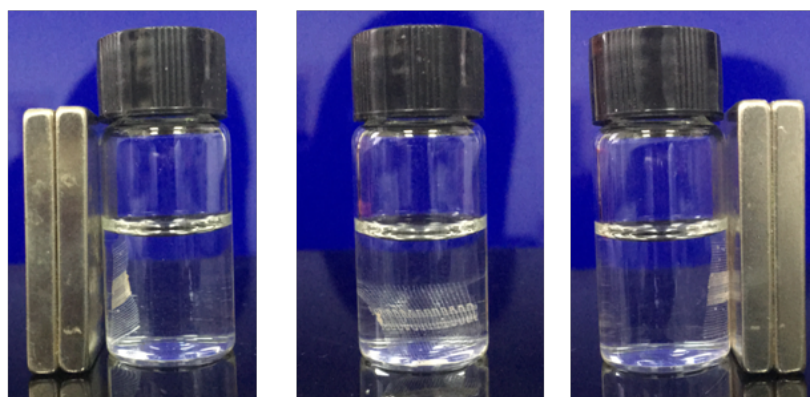
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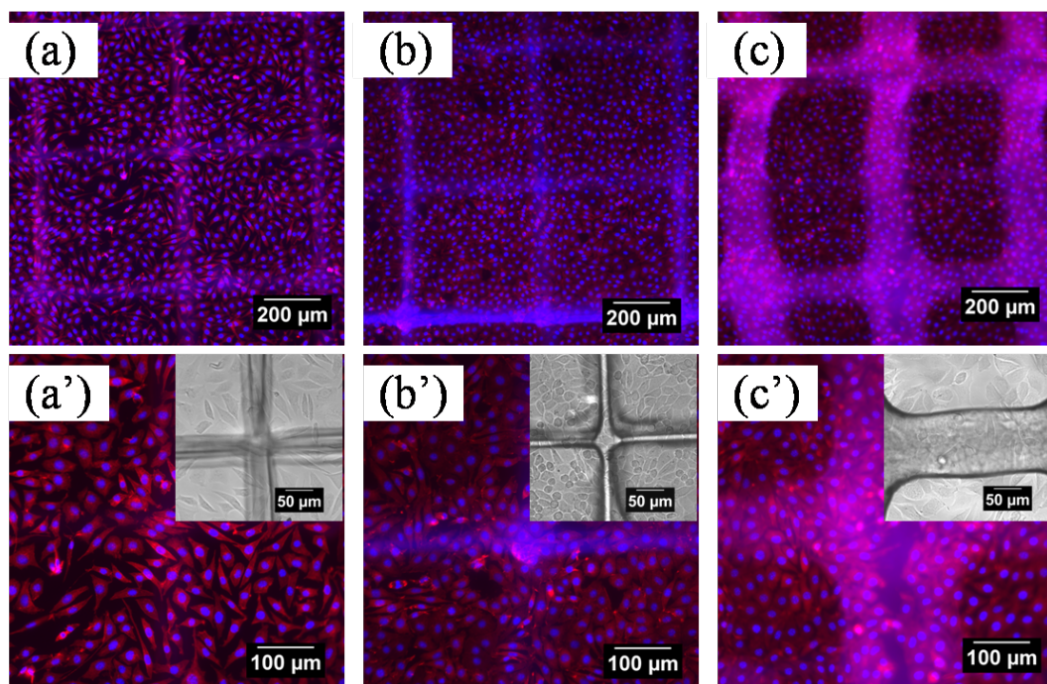
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Figure 10

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Figure 11